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Structural and Dielectric Properties of $Cu_x Zn_{1-x} Fe_2O_4$ (x = 0.2, 0.6, 0.8) Nano Particles

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Abstract

Researchers widely investigate multi-ferrite nanoparticles due to their fascinating magnetic and electrical properties with satisfactory thermal and chemical stabilities. In the present work $Cu_xZn_{1,x}Fe_2O_4(x = 0.2, 0.6, 0.8)$ were synthesized using the auto combustion method. The spinel structure of the prepared samples was verified using XRD. The compositional dependent dielectric and ac conductivity studies were performed using impedance spectroscopy technique. The dielectric properties, such as complex dielectric constant and impedance, have been studied as a function of frequency. Changes of dielectric loss tangent (tan δ) with the frequency have been studied to get information about the energy dispersed inside the materials. The ac conduction study, as a function of frequency, suggests the hopping conduction mechanism at the higher frequencies. From the complex impedance spectra (Nyquist plots or Cole-Cole plots), On the real axis, we identified a dispersion as opposed to a centered semicircle. This suggests a relaxation type other than Debye. The dielectric dispersion observed at lower frequencies can be explained using Koop's phenomenological theory. Since many gases are released during mining and the investigated $Cu_2Fe_2O_4$ is known to be an excellent gas sensor, this study helps to use it effectively in the mining sector.

Keywords: Auto Combustion, Dielectric, Ferrites, Mining, Nyquist Plot

1.0 Introduction

Spinel structure nano ferrites, with chemical formula MFe_2O_4 , have intriguing magnetic and electric properties along with acceptable thermal and chemical stabilities, they are found to play a pivotal role in technical fields such as communication, medical, and electric fields¹. Correspondingly multi-ferrite nanoparticles are widely investigated by researchers². The mentioned properties

are influenced by grain structure, inhomogeneities, presence of voids, surface layers and mainly occupancy of cations in the spinel structure³. These factors can be doctored through various aspects such as synthesize methods, sintering conditions and nature of divalent element⁴. It's noteworthy to know that ferrites can be synthesized through techniques such as co-precipitation, micro-emulsion, solid state reaction, sol-gel etc.

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Spinel ferrite unit cell consists of 8 *fcc* cells of oxygen atom, containing 32 oxygen atoms, 16 iron and 8 divalent cations⁵. There are tetrahedral as well as octahedral sites in a spinel ferrite. Cation occupancy in these sites determine the type of spinel such as inverse, normal and mixed spinels. In inverse spinels Fe^{3+} ions occupy equally among tetrahedral and octahedral sites. In normal spinels Fe^{3+} ions occupy octahedral sites and divalent cation occupy tetrahedral sites. In mixed spinels divalent cations are distributed among both sites⁶. Zinc ferrite is known to exist as normal and copper ferrite crystallizes in inverse spinel structure⁷.

One of the most serious environmental issues affecting the mining sector today is acid mine drainage, or AMD. The ground surface water that drains from a mining site where, sulfide minerals are treated is known as AMD. Sulfuric acid is created when sulfide minerals, especially pyrite and pyrrhotite, oxidize in water when exposed to oxygen. Other minerals are then leached by the acid⁸. Because of this, depending on the location, AMD has a range of metal pollutants. Wang et al⁹. has demonstrated that under ambient temperature conditions acid drainage solutions can be used to make spinel ferrites. The ability of the ferrite process to scavenge the majority of divalent metal ions and the precipitates that are produced to be stable and easily recovered by magnetic filtration are its two main benefits when treating AMD. A two-step continuous ferrite flow method for steady extraction from an actual AMD comprising copper, zinc, and arsenic was presented by Toshifumi et al¹⁰. Another reason to pursue the process is the potential commercial use of ferrite as recording media and magnetic indicators.

Dielectric properties refer to a material's ability to respond to an electric field. These properties are often characterized by parameters such as dielectric constant, dielectric loss, and electrical conductivity. In the case of $CuFe_2O_4$, its dielectric properties can be influenced by factors such as composition, crystal structure, and temperature. On the other hand, in underground mining, the air quality can be compromised by dust, fumes, and gases released during the extraction process. Gas sensors can monitor the air quality in real-time, providing information on pollutant levels and helps to implement measures to maintain a healthy working environment for miners^{11,12}. Gas sensing involves the ability of a material to detect and respond to the presence of specific gases. The interaction between a gas and the sensing material leads to changes in electrical, optical, or other measurable properties. In gas sensing applications, $CuFe_2O_4$ has been explored for its response to various gases, including reducing gases like hydrogen and methane. The interaction between $CuFe_2O_4$ and gases can modulate its dielectric constant. Changes in the dielectric constant may occur as a result of gas-induced alterations in the electronic structure, charge distribution, or polarization of the material. This modulation can be exploited for gas sensing applications¹³⁻¹⁵.

Besides this spinel ferrites cater for the applications of dielectric material owing to the tunability of the materials for the desired region of frequencies. Ferrites can be highly suitable for fabricating cores of transformers, high frequency applications and microwave absorption owing to its minimal dielectric loss and conductivity⁷. Hence engineering the material composition and tailoring the properties of the materials under study is important in knowing their suitability for the desired applications. In particular, dielectric behavior is one of the significant areas of ferrites to be studied due to their dependency on preparation conditions and the type and quantity of dopants¹⁶. Accordingly, attempt has been in this current research work to study the dielectric properties of as prepared Cu_{0.2}Zn_{0.8}Fe₂O₄, Cu_{0.6}Zn_{0.4}Fe₂O₄ and $Cu_{0,s}Zn_{0,2}Fe_{2}O_{4}$ samples.

2.0 Experimental Methods

2.1 Sample Preparation and XRD Studies

The samples were prepared through an auto combustion route. Nitrates of the required metals, copper, zinc and iron were weighted stoichiometrically and mixed with citric acid in deionized water. The solution was then transferred to a preheated furnace at 450° C in a crucible for combustion process. After combustion a foamy powder was obtained. The as obtained sample was grinded to fine powder and further calcinated at 1000° C. X-Ray Diffraction (XRD) plots of the samples were obtained using Rigaku Smart lab to verify crystal formation and phase purity.

2.2 Impedance Spectroscopy Studies

The electrical impedance analysis is carried out by applying an alternating sinusoidal voltage to the sample that is investigated and measuring the *ac* current response. The investigated powder sample is first converted into a pellet by applying hydraulic pressure of 5 tons, and then silver paste contacts are made to both circular cross-sections of the pellet, and copper wires are attached to this silver paste contacts. Silver is chosen in making the contacts due to its high conductivity. These contacts are then kept for annealing at 100°C in the oven for ~12 hours. In the present study, Agilent 4294A impedance analyzer, which is based on the auto-balanced bridge Implementation, is used. The samples under study were scanned from 40 Hz to 20 MHz at room temperature (300 K).

3.0 Results and Discussion

3.1 XRD Studies

XRD diffraction pattern of $Cu_{0.2}Zn_{0.8}Fe_2O_4$, $Cu_{0.6}Zn_{0.4}Fe_2O_4$ and $Cu_{0.8}Zn_{0.2}Fe_2O_4$ are presented in Figure 1. Peaks with planes (220), (311), (222), (400), (422), (511), (440) were identified correspondingly at 2θ positions 29.9, 35.2, 36.8, 42.9, 53.2, 56.7, and at 62.3 according to ICSD database with reference code 01-077-0012 of copper zinc iron oxide. This corresponds to cubic structure with space group $Fd-3m^{17}$. The crystallite size of the samples



Figure 1. XRD spectra of $Cu_x Zn_{1-x} Fe_2 O_4 (x = 0.2, 0.6, 0.8)$ samples.

was estimated using the Debye Scherrer method and was found to be in the range 37 to 47 nm.

3.2 Complex Dielectric Constant

Unlike in a vacuum, the frequency of external fields greatly influences how typical materials react to them. This frequency dependence highlights the fact that the polarization of a material does not change instantly when an electric field is applied. A phase difference can be used to represent the response, which must always be correlational (arising after the applied field). As a result, permittivity is frequently modeled as a complex function of the applied field's (angular) frequency. The complex dielectric constant can be written as a sum of real and imaginary part as given by

$$\varepsilon^*(\omega) = \varepsilon'(\omega) + i\varepsilon''(\omega)$$
 (1)

Both real and imaginary parts of the complex dielectric constant were calculated as per the equations shown below for the complex impedance¹⁸.

$$\varepsilon' = \frac{t}{A\omega\varepsilon_0} \frac{Z'}{Z'^2 + Z''^2} \tag{2}$$

$$\varepsilon'' = \frac{t}{A\omega\varepsilon_0} \frac{Z''}{Z'^2 + Z''^2} \tag{3}$$

Where t is the thickness of the pellet and A is the crosssectional area of the pellet. Z' and Z"correspond to real and imaginary parts of impedance. The variation of both the real and imaginary parts of the complex dielectric constant with the frequency are shown in Figure 2.

Figure 2(a) shows the variation of the real part of dielectric constant ($\dot{\epsilon}$) of Cu_xZn_{1-x}Fe₂O₄ (x = 0.2, 0.6, 0.8) as a function of frequency. From the obtained plots we can clearly see that the values of dielectric constant are higher at lower frequencies and lower at higher frequencies. Maxwell type of polarization phenomena as well as Koop's phenomenological theory, which are in agreement with each other, can explain the variation of dielectric constant¹⁹. According to this theory, exchange of electrons between Fe³⁺ – Fe²⁺ ascertain the applied electric field aiding polarization at lower frequencies. This leads to the variation of the values of dielectric constant ranging from higher value and slowly decreasing as the frequency is advancing. Higher frequencies, on the other



Figure 2. Variation of (a) dielectric constant (ε ') as a function of log(*f*), (b) dielectric permittivity (ε '') as a function of log (*f*) for the systems Cu_xZn_{1-x}Fe₂O₄(x = 0.2, 0.6, 0.8).

hand, prevent the electron exchange between Fe^{3+} and Fe^{2+} from tailing the applied electric field, which results in a lower dielectric constant value. Similar behavior and response we track for the variation of dielectric permittivity as a function of frequency.

3.3 Dielectric Loss Tangent (tan δ) and Impedance Measurements

The information about the energy dispersed inside materials as a result of impurities and defects is given by the dielectric loss tangent. The variation in stoichiometry, the Fe³⁺ content, and structural homogeneity all would affect the dielectric loss tangent. The loss tangent is additionally influenced by the conduction mechanism²⁰.



Figure 3. (a) The variation of dielectric loss tangent $\tan(\delta)$ (b) Variation of real part of impedance (Z') of $\operatorname{Cu}_{x}\operatorname{Zn}_{1-x}\operatorname{Fe}_{2}\operatorname{O}_{4}$ (*x*=0.2, 0.6, 0.8) as a function of frequency.

The decrease in energy observed in the ferrite is measured by the dielectric loss tangent, which is calculated using the formula,

$$\tan\left(\delta\right) = \frac{\varepsilon}{\varepsilon'} \tag{4}$$

The polarization trails the applied alternating field as the dielectric loss tangent grows. The graph of dielectric loss that varies with frequency is shown in Figure 3(a). The experimental finding shows that the dielectric loss factor is high for all the samples (studied in this work) in the lower frequency range. The existence of grain boundaries influences the dielectric loss. Due to the presence of grain boundaries, there is greater resistance, which causes significant dielectric loss. Transport of electrons between Fe³⁺ and Fe²⁺ ions require sufficient energy at lower frequency, resulting in significant dielectric loss. However, because of grains, the resistivity decreases in high frequency regions²¹. The electron transport mechanism between the two iron ions requires only a little amount of energy at the octahedral sites.

Impedance spectroscopy connects the material's microstructures to its dielectric characteristics. Additionally, it aids in the analysis of the effects of numerous elements on polycrystalline materials, such as grain borders, interfaces, or grains. Impedance measurements statistically provide data concerning resistive and reactive components²². Figure 3b, the graph that represents real part impedance, shows fluctuation in the data as a function of frequency. The outcome of the experiment demonstrates that z' decreases with frequency.

3.4 AC Conductivity

The charge carrier and conduction mechanism in ferrite materials are explained by conductivity studies. The most frequent aspects affecting a material's conduction behavior include grain boundaries, mobility of charge carriers, space charge polarization, electron hopping, and particle size. The *ac* conductivity (σ_{ac}) is calculated using the following equation¹⁸.

$$\sigma_{ac} = \frac{t}{A} \frac{Z'}{Z'^2 + Z''^2} \tag{5}$$

Total conductivity σ_{tot} can be written as sum of σ_{dc} and σ_{ac} where the *dc* conductivity is, obviously, a frequency independent part of the total conductivity and is dominant in low frequency ranges, *ac* conductivity follows Jonscher's power law relation, and shows dispersive behavior at higher frequencies.

$$\sigma_{ac} = A' f^s \tag{6}$$

Eqn. for σ_{ac} is the Jonscher's power law²³. is the polarizability strength (temperature independent factor) and 's' represents exponent term and is temperature dependent factor, which gives significant information related to the theoretical model used to explain the conduction mechanism. The variation of *ac* conductivity



Figure 4. (a) Variation of σ_{ac} vs log (*f*) (b) variation of log (σ_{ac}) vs log (*f*) is given for samples Cu_x Zn_{1-x} Fe₂ O₄(x=0.2, 0.6, 0.8). The observed values of *s* are less than 1 supporting hopping mechanism.

 (σ_{ac}) as a function of frequency of $\text{Cu}_x \text{Zn}_{1,x} \text{Fe}_2 \text{O}_4(x = 0.2, 0.6, 0.8)$ is shown in Figure 4. From the Figure it is evident that conductivity slowly increases at the low frequency region, whereas at the higher frequency region, conduction starts increasing exponentially, which could be attributed to the hopping of infinite charge carriers.

Koop's phenomenological theory²⁴ explains the variation in conductivity as a function of frequency quite well. This theory not only explains the mechanism of higher frequency conductivity, but also sheds light on the material aspects. Grain boundaries and grains make up a material or compound. The material's high conductivity is attributable to grains, while its low conductivity is directly

attributable to grain boundaries. At lower frequencies, grain boundaries cope with the applied frequency, resulting in higher resistance and constant conductivity. However, at higher frequencies, grain boundaries have less of an effect and grains aid in conductivity. As a result, conductivity increases exponentially at higher frequencies. In this case, charge carriers hop between adjacent sites, resulting in increased conductivity. Thus, the highly resistive grain boundaries become irrelevant at higher frequencies, and hence the conductivity increases.

3.5 Cole-Cole Plot (Nyquist Plots)

Complex impedance spectrum (Cole-Cole) plots of the samples are demonstrated in Figure 5. From these spectra we can acquire information regarding conduction mechanism and charge transport properties of the materials under study. This indicates a non-Debye type of relaxation since we see dispersion on the real axis unlike a centered semicircle²⁵. According to previous work on ferrites²⁶, researchers have observed two distinct semi circles related to compounds under study. The first one, at low frequency region, is associated with resistance due to grain boundary, whereas one observed at high frequency refers to resistance due to grains. Impedance spectroscopic study is able to distinguish this both because relaxation times of grains and grain boundaries are distinct³. In the present investigation, we have observed



Figure 5. Cole-Cole plot of $Cu_x Zn_{1-x} Fe_2O_4$ (x = 0.2, 0.6, 0.8) samples at room temperature.

an individual semicircle minor arc, demonstrating conduction due to grains is dominant in the systems copper doped zinc ferrites. Since all the measurements are done at room temperature in the present study, they may not necessarily be well resolved. Moreover, in grain boundaries transport properties are also associated with presence of imperfections which could be more predominant than grains.

4.0 Conclusion

 $Cu_Z Zn_{1}Fe_2 O_4 (x = 0.2, 0.6, 0.8)$ samples were successfully prepared through auto combustion route. From XRD analysis homogeneous phase formation of spinel structure with space group Fd-3m was verified. The compositional dependence of electrical parameters was investigated using impedance spectroscopy technique. The observed dielectric behavior is in agreement with Koop's theory and loss tangent impedance variance are concluded to be due to exchange interaction, presence of grains and interfaces. In complex impedance spectrum a non-Debye type of relaxation was observed showing conduction is mostly mediated by grains. The practical deployment of Zn-doped CuFe₂O₄-based gas sensors in the mining industry will greatly increase safety, lessen environmental effect, and boost overall mining operations efficiency. This shift in dielectric characteristics can be utilized for this purpose.

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