

Physical and Optical Properties of Silicon Doped LaOF: Gd³⁺ Nanoparticles Prepared by Combustion Method

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Abstract

The bandgap energy and density of the sample were found by the analytical method. The density was found to increase up to 5 mol%, depicting the compactness of the sample and then decreasing with gadolinium content. Molar volume (V_m), Gd³⁺ ion concentration (N), refractive index (n), electronic polarizability (α_e), reflection loss (R_p), polaron radius (r_p), internuclear distance (r_i), field strength (F), dielectric constant (ϵ) of the sample were calculated.

Keywords: Combustion, Gd³⁺, LaOF, Optical, Physical, Rare Earth, Silicon

1.0 Introduction

In recent days, increasing the development of the white Light-Emitting Diode (LED) industry, rare earth (Gd³⁺) ion-activated lanthanum oxyfluoride inorganic luminescent phosphor materials have been diverse applications in potential fields over the earlier decades, such as White Light Emitting Diodes (WLEDs), cathode ray tubes, plasma display panels, photovoltaic cells, fluorescent bulbs, lasers, optical sensors and innumerable energy storage appliances¹⁻⁵. Amongst all the progressive applications of inorganic Nano Phosphors (NPs), WLEDs have been considered for future solid-state lighting resources due to their favorable properties, such as enriched brightness, eco-friendliness, regulatable emission, improved physical, chemical and thermal stability, long lasting strengths, less power consumption

etc. Presently, in the market, commercially available W-LEDs with a blue InGaN chip in combination with a yellow phosphor (YAG:Ce³⁺)⁶ and a group of red, green, and blue phosphors containing Near ultraviolet (N-UV) LED chips were non-existence of red, green constituent and less color representation index. In this regard, the discovery of novel red, green and blue phosphors-converted light-emitting diodes LEDs (pc-LEDs) is paying attention to increasing the prospects for tunable emission through the phosphors transmission and accessibility of simple approaches for both materials design and production of WLEDs⁷.

We report for the first time the utilization of optimized C-S composite SiO₂@ LaOF: Gd³⁺ NPs as a distinctive fluorescent label for the identification of LFPs. For the visualization of LFPs these fluorescent NPs are successful on various surfaces, such as magazine covers, playing

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cards, metallic scales, aluminium foil, and different types of fruits and vegetables. Initially, SiO₂ C-S NPs were synthesized by Stöber method. Later, SiO₂@ LaOF: Eu³⁺, Li⁺ NP coat was developed on SiO₂ by solution combustion method, and all the samples were well characterized to study the physical and optical properties.

2.0 Experimental

The LaOF: Gd³⁺ NPs were fabricated using the solution combustion method, and all the chemicals used were of analytical grade with 99.9% purity (Sigma Aldrich). According to stoichiometric calculation, Lanthanum nitrate [La (NO₃)₃.6H₂O], Ammonium fluoride [NH₄F] and Gadolinium nitrate [Gd (NO₃)₃.6H₂O] were used as starting materials and EGCG extract as a fuel. These mixtures were run through deionized water in a Petri dish and make a homogeneous mixture by using a magnetic stirrer for ~20 min. This clear solution was kept in a preheated muffle furnace (500 ± 10 °C). The solution changed into lotion and then became burned. The final product was calcinated for 2 h at ~ 800 °C. The schematic illustration for the synthesis of LaOF: Gd³⁺ NPs.

3.0 Results and Discussion

3.1 Density and Molar Volume

The density of the LaOF sample was found using Archimedes Principle. Acetone (0.786 g/cm³) was used as an immersion liquid for the sample to measure its density. The sample density was calculated using the relation.

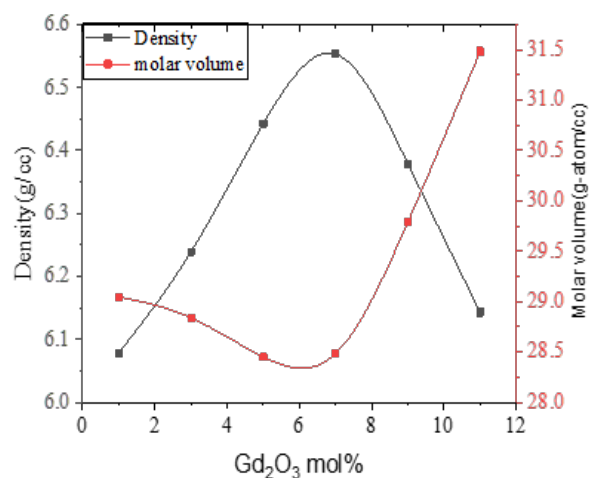


Figure 1. Variation of density and molar volume with Gd₂O₃ content.

$$\rho_{glass} = \frac{W_A}{W_A - W_B} 0.786 \text{ (g/cm}^3\text{)}$$

where, W_A is the weight of the sample in air and W_B is the weight of the sample in acetone. The measurement was repeated three times for each sample to approximate the average value. The values are listed in Table 1.

The density for the sample increases, and the corresponding molar volume decreases up to 7 mol% gadolinium content due to bridging oxygen atoms and then finds the opposite nature beyond 7 mol% as shown in Figure 1 due to non-bridging oxygen atoms. The non-linear variation of density, molar volume and Oxygen

Table 1. Density (ρ), Molar volume (V_m) and Oxygen Packing Density (OPD) with Gd₂O₃ mol%

Phosphor	Density (g/cc)	Molar volume (x 10 ⁻⁵ m ³)	OPD (g-atom/litre)
LaOF:Gd ³⁺ (1 mol%)	6.078	29.080	35.075
LaOF:Gd ³⁺ (3 mol%)	6.239	28.932	36.636
LaOF:Gd ³⁺ (5 mol%)	6.442	28.601	38.459
LaOF:Gd ³⁺ (7 mol%)	6.553	28.691	39.733
LaOF:Gd ³⁺ (9 mol%)	6.378	30.063	39.250
LaOF:Gd ³⁺ (11 mol%)	6.143	31.829	38.329

Packing Density (OPD) with Gd³⁺ concentration is due to the electronic and ionic hopping mechanism in the network⁸.

3.2 Physical and Optical Properties

Gd³⁺ ion concentration (N), density (ρ) and refractive index (n) were used to estimate the physical and optical properties of the samples, such as electronic polarizability (α_e), reflection loss (R_L), polaron radius (r_p), inter nuclear distance (r_i), field strength (F) and dielectric constant (ϵ) using the standard relations and the obtained values are listed in Table 2.

3.2.1 Ion Concentration (N)

Gadolinium ion concentration (N) can be calculated using the relation⁹

$$N = \frac{(\text{Avogadro's number}) (\text{glass density})}{(\text{glass average molecular weight})} (\% \text{ of RE})$$

The concentration of Gd³⁺ ions increases in the sample with the percentage of Gd³⁺ ions. Because of ratifying the compactness of the system with the addition of gadolinium oxide, the polaron radius and inter nuclear distance has been calculated using the below relations¹⁰.

$$\text{Polaron radius, } r_p = \left(\frac{1}{2}\right) \left(\frac{\pi}{6N}\right)^{1/3}$$

$$\text{Inter nuclear distance, } r_i = \left(\frac{1}{N}\right)^{1/3}$$

where, N indicates Gd³⁺ ion concentration. Table 2 depicts a decrease in inter nuclear distance and polaron radius with Gd³⁺ ion concentration. This result leads to the Gd-O bond strength increases, which enables the strong field near the Gd³⁺ ions.

The compactness of the sample shows that the degree of delocalization of electrons decreases, which results in a decrease in band gap energy^{11,12}.

3.2.2 Field Strength (F)

Field strength gives information about the arrangement of ions in the sample the values were calculated using the relation given¹⁰ and were presented in Table 2.

$$\text{Field strength, } F = \left(\frac{z}{r_p^2}\right)$$

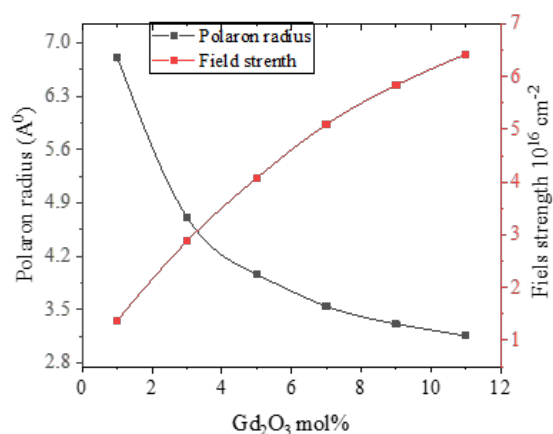


Figure 2. Variation of polaron radius and field strength with Gd₂O₃ mol%.

where, Z is the atomic mass of gadolinium oxide and r_p is the polaron radius.

The obtained values show the rigidity of the sample increases with Gd³⁺ content. The variation of polaron radius and field strength with Gd₂O₃ content is shown in Figure 2. This figure shows field strength increases while polaron radius decreases. In general, field strength and polaron radius shows the opposite behavior and which is observed in our study.

3.2.3 Refractive Index and Electronic Polarizability

When light interacts with atoms of the sample, the parameter refractive index can be calculated, and this parameter depends on cation polarizability and density. The electronic polarizability (α_m) is used in the field of applications like optical and electronic devices¹³. Due to the application of the electromagnetic field, the electronic polarizability exhibits distortion in the electronic cloud¹⁴. The parameter α_m was measured using the Clausius-Mosotti equation

$$\alpha_m = \left(\frac{3}{4\pi N_A}\right) R_m$$

where, R_m is the molar refraction and N_A is Avogadro's number

$$\text{The molar refraction } R_m = \left(\frac{n^2 - 1}{n^2 + 2}\right) V_m$$

where, V_m is the molar volume, and n is the refractive index

The variation of refractive index and electronic polarizability is shown in Figure 3. From the figure the increasing values may be due to increasing in Gd^{3+} ion content, which has high electronic polarizability ($7.432 \text{ (Å}^0)^3$) than B^{3+} ($0.002 \text{ (Å}^0)^3$)^{15,16}.

3.3.5 Metallization Criterion(M)

The metallic and non-metallic nature of the sample was based on metallization criterion values¹⁷. The metallization criterion of the sample can be calculated using the relation given below

$$M = 1 - \frac{R_m}{V_m}$$

where, R_m is molar refraction and V_m molar volume.

If the value $R_m/V_m < 1$, the sample is non-metallic and if the value $R_m/V_m > 1$, the sample is metallic. It is established that from the above equation, the sample is non-metallic, and the calculated values are listed in Table 2. The highest value of the metallization criterion (0.454) calculated for the lanthanum oxyfluoride discloses that lanthanum-ion reveals the highest band gap energy and insulating nature in the sample, which is confirmed by the band gap energy values. Larger metallization criterion values depict that width of both conduction and valence bands reduces, which intern shows a wider optical band gap energy¹⁸.

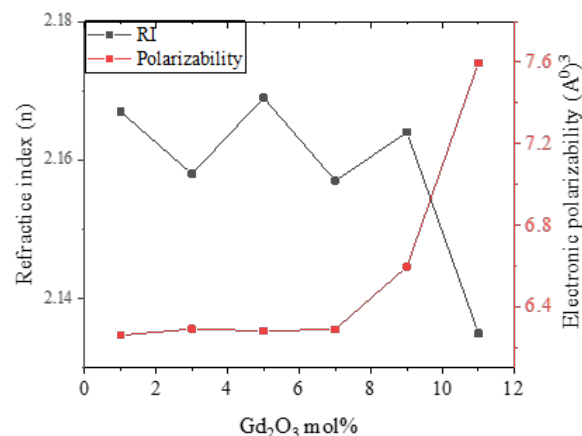


Figure 3. Variation of refractive index and electronic polarizability with Gd_2O_3 content.

3.3.6 Optical Band Gap Energy

The optical band gap energy for the sample is calculated using Davis and Mott theory¹⁹. Optical transitions and electronic band structures were explained because of the absorption effect, which could be able to elaborate the optical band gap energy²⁰. The absorption coefficient $\alpha(\theta)$ is a function of photon energy for indirect and direct band gap energy.

$$\alpha = \alpha_0(h\theta - E_g)^n/h\theta$$

Table 2. The physical and optical properties of $LaOF:Gd^{3+}$ NPs

Physical and optical properties	LaOF:Gd ³⁺ nanophosphors					
	1 mol %	3 mol %	5 mol %	7 mol %	9 mol %	11 mol %
Gd ³⁺ concentration ($\times 10^{20}$ ion/cm ³)	2.071	6.2452	10.529	14.695	18.031	20.815
Polaron radius (r_p) (Å)	6.81	4.714	3.961	3.544	3.311	3.156
Inter nuclear distance (r) (Å)	16.902	11.699	9.829	8.796	8.216	7.832
Field Strength (F)($\times 10^{16}$ cm ⁻²)	1.38	2.88	4.078	5.095	5.839	6.426
Refractive Index (n)	2.167	2.142	2.169	2.157	2.164	2.135
Reflection loss R %	0.135	0.132	0.136	0.134	0.135	0.131
Dielectric constant (ϵ)	4.697	4.589	4.708	4.655	4.683	4.562
Optical dielectric constant (dpt/dp)	3.697	3.589	3.708	3.655	3.683	3.562
Molar Refraction (R_m) (cm ⁻³)	16.053	15.759	15.81	15.759	16.568	17.278
Metallization criterion	0.447	0.455	0.447	0.45	0.448	0.457
Electronic polarizability (α_m) (Å ³)	6.26	6.291	6.282	6.289	6.597	7.597
OPD (g-atom/litre)	35.075	36.636	38.459	39.732	39.25	38.329

where, E_g optical band gap energy, α_0 is energy independent constant and n is the exponent. If the exponent $n=1/2$ and 2 represents the allowed direct and indirect transition. It is found that band gap energy (E_g) values decrease up to 7mol% of Gd₂O₃ content and then increase as shown in Figure 4, the values are listed in Table 1.

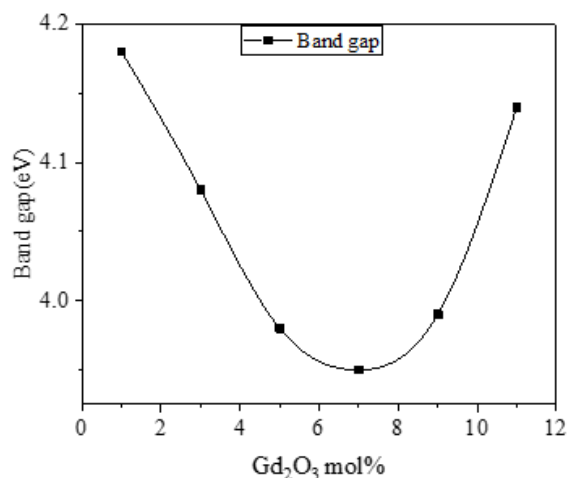


Figure 4. Variation of band gap with Gd₂O₃ content.

4.0 Conclusion

The LaOF: Gd³⁺ NPs were fabricated using the solution combustion method and estimated the physical and optical properties. The density of the samples found non-linear variation due to the bridging and non-bridging of oxygen atoms, which shows the stiffness and rigidity of the samples. The concentration of Gd³⁺ ions increases in the sample with the percentage of Gd³⁺ ions. Because of ratifying the compactness of the system with the addition of gadolinium oxide, the polaron radius and inter nuclear distance has been calculated, and we found that field strength increases while polaron radius decreases.

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